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AFM Evaluation of Different-Sized Active Materials and Interface of All-Solid-State Lithium-Ion Batteries

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Introduction

With recent global warming, improving energy use efficiency with high-performance storage batteries is an important issue. All-solid-state lithium-ion batteries (ASSLiB) with excellent safety, long life, and high energy density have been proposed as a candidate for high-performance batteries.

Problem

The electrode and solid electrolyte form a solid-solid interface, which significantly increases the interface resistance and remarkably lowers the battery performance for the following reasons. ✓ Deterioration of contact state between the electrode active material and solid electrolyte

 \checkmark Depletion of Li⁺

In our previous work, voids were observed at the interface of ASSLiB using TiO₂ with average particles size of 1 μ m as the negative active material. These voids may prevent the movement of Li⁺ and contribute interfacial resistance¹

In this study we investigated ASSLiB using TiO₂ with a particle size of 150 nm with Atomic Force Microscope (AFM) to evaluate the effect of degradation by charge-discharge tests.

Materials & Methods

◆Materials

Oxide-based NASICON (Na Super Ionic Conductor) type ASSLiB \checkmark As assembled : (A) \checkmark After charge-discharge tests : (B)



Fig.1 Optical microscope image of (A)



After the surface of ASSLiB was cross-sectioned, chargedischarge tests were performed. The theoretical capacity of this ASSLiB piece is 150mAh/g, but the actual charge capacity is about half. And the performance was degraded by the charge-discharge tests.(Fig.3).

Fig.3 Charge-discharge characteristic of ASSLiB



✓ Kelvin Probe Force Microscope (KPFM)



✓ Current



Electrical properties of surfaces are observed by detecting the current flowing through the cantilever.

Fig 4. Flow-type glove box (made by Glovebox Japan inc.)

∕ ●Acetylene Black (AB)

•Active materiall •Conductive assistant

Fig.2 Overview of assembled ASSLiB

Surface electric potential is observed by detecting the static electric force acting on the cantilever.

Results & Discussion

•Size evaluation of the negative active materials The shape of the negative active material as assembled was observed.

Different particle sizes of TiO_2 can be clearly seen (Fig.5). When the width of any 4 particles was measured from the topography, the average value was close to the nominal value of the material (Table 1).

 150 nm TiO_{2}





Fig.5 Topography of the negative active materials

Table 1 Size measurement of 150 nm

Average particle sizes [nm]

•Visualization of the conductive path at the positive electrode

The current distribution in 150 nm TiO_2 battery operating conditions was evaluated.

Active material and solid electrolytes have high resistance, so current does not flow very much. Therefore, we can see that the area through which the current flows is the conductive assistant (Fig. 7). From this figure, we can consider the following: \checkmark As assembled and after charge-discharge tests, no deterioration in the current path can be confirmed.

 \checkmark The distribution of the conductive assistants was uneven with respect to the 150nm TiO₂ particles. The conductive path to the active material was insufficient. \checkmark In order to improve the charge/discharge characteristics, it is necessary to improve the dispersion state of the conductive assistant.



Fig.7 Conductivity of positive electrode

 $1 \mu m TiO_2$

t of TiO ₂ particles	
0 nm TiO ₂	1 μm TiO ₂
160.1	1005

•Improvement of the bonding state of the interface

The shape of the negative electrode-solid electrolyte interface as assembled was observed.

There are voids at the interface when the hot press is not used. When hot pressing was used, there were no voids at the interface, which was successfully improved to a dense structure.



•Visualization of the degradation state of positive active material KPFM measurements were performed on the 150nm TiO₂ positive electrode as assembled

and after the tests.

The average potential is 0.75 V before charging and 2.98 V after charging/discharging (Fig. 8). Originally, the potential after discharge should approach 0 V, but the charge remains in this sample. As shown in Fig. 8, there is no deterioration of the electron conduction path, suggesting that a problem occurred in the ion conduction path.



Conclusion

- \checkmark We were able to improve the bonding state of the interface by hot pressing.
- necessary.
- ✓ Degradation caused by charging and discharging can be captured by KPFM measurement.

Reference

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Fig.6 Topography of interface of (A)

Fig.8 Surface potential of the positive electrode

 \checkmark The shapes of different particle sizes of TiO₂ were observed with high resolution by SPM/AFM. \checkmark The conductive assistant was distributed unevenly, and improvement of the dispersion state was

1) E. Iida et al., "SPM/AFM Evaluation of Interface of All-Solid-State Lithium-Ion Batteries", IVC-22,